Review Laser surface alloying: a bibliography

C. W. DRAPER, C. A. EWING

AT & T Technologies, P.O. Box 900, Princeton, New Jersey 08540, USA

Laser alloying is a material processing method which utilizes the high power density available from focused laser sources to melt metal coatings and a portion of the underlying substrate. Since the melting occurs in a very short time and only at the surface, the bulk of the material remains cool, thus serving as an intimate heat sink. Large temperature gradients exists across the boundary between the melted surface region and the underlying solid substrate. The result is rapid self-quenching and resolidification. Quench rates as great as 10¹¹ K sec⁻¹ and concomitant resolidification velocities of 20 m sec⁻¹ have already been realized. What makes laser surface alloying both attractive and interesting is the wide variety of chemical and microstructural states that can be retained because of the rapid quench from the liquid phase. These include chemical profiles where the "alloyed" element is highly concentrated near the atomic surface and decreases in concentration over shallow depths (hundreds of nanometers), and uniform profiles where the concentration is the same throughout the entire melted region. The types of microstructures observed include extended solid solutions, metastable crystalline phases, and metallic glasses. This bibliography is a compilation of published work in this area.

1. Introduction

The use of laser sources to produce metal surface alloys has become a materials processing application receiving considerable attention. The concept of laser surface alloying (LSA) is schematically pictured in the sequential cross-section views in Fig. 1. In Fig. 1a the metal substrate (B) coated with a thin metal film (A) is irradiated with a laser pulse. A fraction of the incident laser light is absorbed by free carriers within the electromagnetic skin depth $(10^{-6} \text{ to } 10^{-5} \text{ cm})$. For metal surfaces and most laser wavelengths a significant fraction of the incident light will be specularly or diffusely scattered away (the reflectance process is not shown in the illustration). The absorbed energy is "instantaneously" (10^{-12} sec) transferred to the lattice. The near surface region very rapidly reaches the melting point and a liquid/ solid interface starts to move (solid arrows) through the film as shown in Fig. 1b. In Fig. 1c the liquid/solid interface has swept through the original thin film/substrate interface. Interdiffusion of the film (solid circles) and substrate (open

circles) elements starts. The laser pulse is nearly terminated and the surface has remained below the vapourization temperature. In Fig. 1d the maximum melt depth has been reached. Interdiffusion continues. The resolidification interface velocity is momentarily zero and then rapidly increases. The resolidification interface then moves approximately half way back to the surface from the melt depth (Fig. 1c). Interdiffusion in the liquid continues but the resolidified metal behind the liquid/solid interface cools so rapidly that solid state diffusion may be neglected. In Fig. 1f the material is completely resolidified and a "surface alloy" of A in B has been produced. What makes laser surface alloying both attractive and interesting is the wide variety of chemical and microstructural states that can be retained because of the rapid quench from the liquid phase.

Fig. 2 is a bar graph of laser surface alloying (LSA) publications as a function of time. Much of the interest is rationalized from the standpoint of strategic element shortages and precious metal costs. All directed energy (laser, electron, and ion



beam) methods of modifying surface composition are by their very nature materials conservation oriented. When an element is incorporated in a bulk alloy solely for the purpose of affecting the



Figure 2 Growth in the LSA literature over the last 20 years.

alloys response to its interface environment, surface rather than bulk alloying is attractive.

Figure 1 Sequential cross-section schematic illustration of mechanism of laser surface alloying from arrival of laser pulse to complete resolidification.

Much of the more recent work explores the use of Q-switched laser sources to achieve high velocity regrowth (10 m sec^{-1}) of the molten metal interface. Extended solid solutions, metastable crystalline and amorphous surface alloys have all been formed by LSA processing. Other reasons for the growing interest are the wider availability of laser sources capable of energy fluences sufficient to melt metal surfaces and some spin-off from the immense push into semiconductor laser annealing.

Whatever the reason, it appears that there will continue to be significant growth in the literature on LSA, and that a timely updated bibliography [108] will aid interested parties in keeping abreast of current research activity.

In assembling Tables I and II, we have attempted to include all scientific literature that falls within the scope of laser surface alloying. Search methods included computer keyword and author searches of chemical, physics, metals, and laser abstracts. Written correspondence and telephone conversations with other active researchers in this field have been very helpful.

Table I is a chronological list of the laser surface alloying literature. No attempts have been made to order the literature on a finer scale than the year of publication. Table headings include the reference year, type of laser sources, alloying specie/host substrate, post-irradation analysis of

TABLE I

Year	Laser	Alloying specie/substrate	Analysis	Reference
1964	P-Ruby	C, W, TiC, WC/St	COMP:EPM	[1]
			SSB:MH	
1969	P-NdG	C/Fe	STRT:XRD	[2]
		-,	SSB:MH	[]
1969	P-Ruby	WC/St	STRT:OM	[3]
1707	2		COMP·FPM	[0]
			SSB·WR	
1972	P-Ruby	C WC/St	STDT-YPD	141
1772	I Ruby	0, 40/51	STRIARD	["]
1074	P-NAC	$C = W/E_{\alpha} \cdot C_{\alpha}/W \cdot A1/Nh$	STDT-OM VDD	[5]
1075	CW-CO	C_r NE W V Mp C/St	COMP: A NC	[5]
1975	C # - C 02	C_1 , M_1 , W , V , M_1 , $C/5t$	SCD-MH	[0]
1076	D NAC	NE MO TE TO MID V/EO St	SSD.MII	171
1077	P NAC	M_{τ} , M		[/]
1977	P-NuG	Mo, Nb/Fe	COMP: ANG	[0]
19//	P-NaG	Mo/Fe	STRT:UM, AKD	[9]
			COMP:XRD	
1070			SSB:MH	6101
1978	P-NdG	V/Fe	STRT:OM, XRD	[10]
			COMP:XRD	
			SSB:MH	
1978	CW-CO ₂	CrC, WC/St	STRT:OM	[11]
			COMP: ANG	
			SSB:MH	
1979	CW-CO ₂	Cu/Ag-7 Cu	STRT:OM, SEM	[12]
1979	Q-Ruby	Cu/A1	STRT:CHN	[13]
			COMP: RBS	
1979	CW-CO ₂	Cr, Ni/St	STRT: OM	[14, 15]
			COMP:EPM, EDX	
1979	CW-CO ₂	Cr-C-Mn-Al/St	STRT:OM	[16, 20]
	-	,	COMP: EPM	
			SSB:MH	
1979	P-NdG	Fe/Nb	STRT: MOSB, XRD	[17]
1979	CW-CO.	$M_0 - N_i - S_i - C_T - C/S_t$	STRT: OM XRD	[18]
	0		COMP.FDX ANG	[10]
			SSB-MH HTS WR	
1979	CW-CO	Cr/St	STRT:OM SEM	[10]
1777		Ci/St	COMPERMENT AFS	[17]
1980	CW-CO O-NdV	Au Ag Pa Sp Ta/Ti	STRT-OM SEM CUN	[21]
1700	$CWCO_2, Q-War$	Au, Ag, 1g, 51, 1a/ 11	COMD DDS EDV AES ESCA	[21]
			COMP. RDS, EDA, AES, ESCA	
1090	CW CO	THO WOLTH CALLAN	SSB:MH, COR	(22)
1960	Cw-CO ₂	$\Pi C, WC/\Pi - 6AI - 4V,$	STRT:OM, SEM	[22]
1080	O NHC	Inconei		
1980	Q-NaG	Sb/Al	STRT:OM, XRD	[23]
1000	0 D /		COMP: RBS	
1980	Q-Ruby	Ni/Fe/SiO ₂	STRT: TEM	[24]
			SSB:MAG, OPT	
1980	CW-CO ₂	TiC, WC/St, Ti–6Al–4V,	STRT:OM	[25]
		Inconel, 5052Al	SSB:MH	
1980	P-NdY	Re/Fe-50 Ni	STRT:OM, SEM	[26]
			COMP: EPM	
1980	Q-NdY	Au, Ag, Ta/Ni	STRT:OM, CHN	[27]
			COMP:RBS	
1980	Q-dNdY	Pb/Cu	STRT:CHN	[28]
			COMP: RBS	
1980	CW-CO ₂	TiC/304 St	STRT:SEM	[29]
	~		SSB:MH	11
1980	Q-Ruby	Cu, Pb/Al	COMP: RBS	[30]
1980	CW-CO ₂	B, FeB/St	STRT:OM. SEM TEM	[31]
	··· = = 2		SSB MH HTS	[21]

TABLE I Continued

Year	Laser	Alloying specie/substrate	Analysis	Reference
1980	CW-CO ₂	Cr/4815 St; Ni, Cr, Si/Al	STRT:OM	[32]
	-	VC, WC/St, TiC/Ti; Ni-CrAl/St	COMP: ANG	
1981	O-Ruby	Zn, Sb/Al	STRT: TEM, CHN	[33]
1701	21000)		COMP:RBS	
10.91	O-NAV	Au Ag Pd Ta Sn/Ni	STRT TEM CHN	[34]
1901	Q-INU I	Au, Ag, 10, 12, 00/10	COMPERS	[]
1001	O Buby	NG/ A.1./NG	STRTITEM	[35]
1901	Q-Kuby	INI/ Au/ INI	COMBADE	[55]
1001	0 D-1	Mr. Cillai		[26]
1901	Q-Kuby	Mo, Cu/Al		[50]
1001	0.000	**C/>*'	COMP.RB5	[27]
1981	Q-Nd Y	HI/Ni	SIRI CHN, PAC	[37]
			COMP:RBS, AES	
1981	Q-NdY	Pd/Ti	COMP:RBS	[38]
			SSB:COR	
1981	CW-CO ₂	TiC, WC, WC–Co/St	COMP:EDAX	[39]
			STRT:SEM	
			SSB:MH	
1981	CW-CO,	Cr. Cr–Co/St	STRT:OM	[40]
	- 2		SSB:MH	
1981	CM-CO. O-NdY	Au/Ni	STRT:CHN	[41]
1701	O_{2} , Q Ruly, O_{2} , Q Ruly	1100/111	COMP'RBS	(·-)
1981	$CW_{CO} = O_{NdV}$	A 11 / Nii	STRTCHN	[42]
1701	O dNdV	Aujin	COMPORES	[]
1001	Q-uNu I O Buby	$C_{\rm T}$ Sb/A1	TUDV-UE MT	[43]
1901	Q-Ruby	CI, SD/AI	THE LOF, MI	[45]
1981	Q-aNa Y	Ag/ Ni	SIRI:CHN	[44]
	ann a a		COMP:RBS	1471
1981	CW-CO ₂	TiC/Ti-6Al-4V, 304 St,	STRT:OM, SEM	[45]
		4340 St, 5052 Al, Al-bronze		
1981	CW-CO ₂	Si, TiC/5052 Al	STRT:OM, SEM	[46]
1981	CW-CO ₂	Fe-B/Fe-12Cr-2C,	STRT:TEM, OM	[47]
		Ni ₆₀ Nb ₄₀ /Nb	SSB:MH, HTS	
1981	CW-CO ₂	Cr/St	SSB:COR	[48]
1981	Q-NdG	Cr/Al	COMP:RBS, SEM	[49]
1981	CW-CO,	Cr, Cr-Ni/St	COMP:ANG	[50]
	-	Cr-W-TiC/St	SSB:MH	
1981	CW-CO.	TiC. $WC/Ti-6Al-4V$,	STRT:SEM	[51]
	0.1.002	304 St. 5052 Al	SSB:MH, WR	
1981	O-Ruby	In/Al	STRT:PAC	[52]
1981	CW-CO	Cr/Fe	COMP:EPM	[53]
1701		CITE	STRTOM	[]
10.91	O NAC	Sh/A1	COMP'RBS_SEM	[54 55]
1001	Q-NuO Q Bubu	$\frac{7n}{A1}$ Au/Ni	THRV-MT	[56]
1902	Q-Kuby Q NdC	Db/A1	COMP: BBS_SEM	[57]
1902	Q-NuG		Sorat coaff	[57]
1982	Q-Ruby	Cu/AI	Solet coell.	[50]
1982	Q-Ruby	V – Si/sapphire	COMP: KBS	[39]
			SIRI:ARD	
			SSB:SC	[(0)]
1982	Q-Ruby	Cr/Al	COMP:RBS	[60]
			STRT: CHN, SEM	
1982	Q-Ruby	Cr, Mo/Al	COMP:RBS	[61]
			STRT:CHN	
			SSB:HTS	
1982	Q-Ruby	Eu, Ni/Ni	COMP:RBS	[62]
			STRT:CHN, SEM	
1982	Q-Ruby	Sb/Al	STRT:CHN, TEM	[63]
1982	Q-dNdY	Cr, Ni/Cu	COMP: RBS, AES	[64]
	× · · · ·	<i>,</i> ,	STRT:CHN, TEM, XRD	
			SSB:COR	
1982	O-dNdY	ZI/NI, Al, Fe, Ti, V	COMP:RBS	[65]
1200	×	Cu/Zr	STRT: TEM, CHN, XRD	
		· · · · · · · · · · · · · · · · · · ·		

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Year	Laser	Alloying specie/substrate	Analysis	Reference
1982	Q-Ruby	Cu, Zn/Al	THRY:MT	[66]
1087	CWCO	Au/Ni WC Co WC Co Mo	COMPLANC	[67]
1702		WC = Ni = Cr = B = Si/St	STRT:OM SEM	[07]
		we-m-er-b-b/st	SSR·MH	
1982	CW-CO	Fe_B/St	STRT-VRD TEM	[68]
1982	$CW-CO_2$	$C_r/F_{e} = 0.2C$	COMPEDAY	[00]
1702		CI/1'e=0.2C	STRT-OM TEM	[09]
1982	CW-CO	Eati Eacr Each EaV Easi/St	STRI.OM, IEM	[70]
1982		8×1 12Mo 3×1 Ni	COMP. ANC	[70]
1702		$12A1 + 15M_{\odot} = N;$	COMF. ANG STRT-TEM	[/1]
		12AI-15MO-NI	SIKILLM	
1987	CW-CO	Mo/St	SSD. INS COMPEDAY	[72]
1702	C = C = C + C + C + C + C + C + C + C +	MO/St	STDT-SEM	[72]
			STRT.SEM SCR·MU	
1982	CW-CO	Ni/Fe	COMDEDM	[72]
1702	$C_{11}C_{2}$	NI/ I C	THEVIME	[75]
1982	O-N4G	$S_{\rm m}/A1$ W/V Db/Eq. W/Cr		[74]
1902	Q-Mag	SII/AI, W/V, PO/Fe, W/CF	COMP:RB5	[/4]
1003	CWCO	0. /F-	SIRI;CHN	[2.5]
1982		Cr/Fe	COMP:EPM	[75]
1002	0.000		STRT:SEM	19 (1)
1982	Q-NdG	Pb/Al	COMP:RBS	[76]
1003			STRT:SEM	
1982	ML-NdY	Fe ₃ B/Fe	STRT:TEM	[77]
1982	CW-CO ₂	Al, Cr, Ni/Fe	COMP: EPM	[78]
			THRY:MT	
1982	CW-CO ₂	Ni/Fe	COMP:EDAX	[79]
			STRT:TEM, OM	
1982	CW-CO ₂	Cr/St	COMP:EDAX	[80]
			STRT:TEM, SEM	
			SSB:MH	
1982	CW-CO ₂	Cr-Ni/4140 St	COMP: EPM	[81]
			SSB:COR	
1983	Q-Ruby	Cu/Au/Co	COMP:RBS	[82]
		Cu/W/Cu		
1983	Q-NdY	Hf/Ni	COMP:RBS	[83]
			STRT:PAC, CHN	
1983	CW-CO ₂	Mo/6150 St	COMP:EDAX	[84]
			STRT:SEM, TEM	
			SSB:MH	
1983	CW-CO ₂	Ni/1020 St	THRY:MT	[85, 88]
		Cr/1018 St	COMP:EPM	
1983	CW-CO ₂	Cr, Ni/Fe;	COMP:EDAX	[86]
		Cr/Ni/Fe	STRT:OM, SEM, TEM	
1983	CW-CO ₂	Cr/Fe	COMP:EDAX	[87]
			STRT:OM, SEM, TEM	
			SSB:MH	
1983	ML-NdY	Fe ₃ B/Fe	STRT: TEM	[89]
1983	P-CO ₂	Ar, N_2 , CO_2 , C_2H_4/St	STRT:OM	[90]
			SSB:MH	
1983	Q-Ruby	N, C/Mo	COMP:RBS	[91]
			SSB:SC	
1983	Q-NdG	Ni ⁶² /Ni, Fe ⁵⁵ -Fe ⁵⁹ /Fe	COMP:RI	[92]
			THRY:MT	. ,
1983	CW-CO ₂	Cr/St	STRT:OM, SEM, TEM	[93]
			SSB:MH	. ,
1983	CW-CO ₂	Cr, WC-10Co/St	STRT:OM	[94]
			SSB:MH	1 J
1983	CW-CO ₂	TiC/6061 Al	STRT:SEM	[95]
			SSB:WR	E 1

TABLE I Continued

Year	Laser	Alloying specie/substrate	Analysis	Reference
1983	EB	Ag/Al, Mn/Al, As/Si	THRY:MT	[96]
1983	ML-NdY	Fe ₃ B/Fe	STRT:TEM	[97]
1984	ML-NdY	Fe ₃ B/Fe, Ni/Nb	STRT: TEM	[98]
		Mo/Ni, Mo/Co, Nb/Co		
1984	CW-CO ₂	Cr/St	COMP:EDAX	[99]
			STRT:SEM, TEM	
1984	Q-Ruby	La, Eu/Ni	COMP:RBS	[100]
			STRT:CHN	
1984	Q-Ruby	Pb/Ni	COMP:RBS	[101]
			STRT:SEM	-
1984	Q-Ruby	Au, Co, Cr, Zr/Ti	STRT:XRD	[102]
1984	Q-dNdY	Cu/Zr	STRT:XRD, TEM	[103]
1984	Q-dNdY	Ni-8V/Cr/Cu-2Sn	COMP: AES, RBS	[104]
1984	CW-CO ₂	Cr/St	COMP:EDAX	[105]
	-		STRT:OM	
			SSB:MH	

structure, composition or modification of surface sensitive behaviour, and finally the numerical reference. Abbreviations used are explained in Table II. Take for example the 1978 Kovalenko work [10]; samples of iron coated with vanadium were alloyed with a pulsed neodymium:glass laser. Alloyed samples were analysed structurally using optical microscopy and X-ray diffraction. Compositions are given based on lattice parameter changes in the X-ray diffraction patterns. The microhardness of the irradiated sample is reported.

Table I does not provide insight into how the alloying elements were put down prior to laser irradiation. The methods used are numerous and include vacuum evaporation, electroplating, electroless plating, powder coatings, thin-foil application, and ion implantation. In some cases the element is injected into the melt region in the form of particles or wire. These last two methods of introducing the alloying specie are the most attractive from a commercial standpoint. When coupled with repetitive laser scanning it has been demonstrated that one can build up the surface alloyed region. This is the basis of the United Technologies effort in a commercial processing method called "layer glazing". A sizable body of that work has been summarized [112, 113].

There have been several short review type articles which may be useful to the reader interested

TABLE II	Abbreviations	used	in	Table	I
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Laser		
P, pulsed		
CW, continuous		
Q, Q-switched		
ML, mode locked		
CO_2 , carbon dioxide		
NdY, neodymium yttrium aluminium garnet		
NdG, neodymium glass		
dNdY, frequency doubled NdY		
Ruby, chromium doped aluminium oxide		
EM, electron beam		
EX, excimer		
Alloying specie/substrate – elemental symbols		

Alloying specie/substrate – elemental symbols St, steel, cast irons, general designation for Fe alloy substrates Commas (,), denote individual experiments Dashes (–), denote a mixture of elements Semicolon (,), denote individual experiments (added clarity)

For example, C, WC/St denotes carbon film and tungsten carbide film on a steel substrate Re/Fe-50Ni denotes a rhenium film on an iron-50 wt % nickel alloy substrate Analysis STRT, structure SEM, scanning electron microscopy TEM, transmission electron microscopy OM, optical microscopy XRD, X-ray diffraction CHN, channelling MOSB, Mössbauer spectroscopy PAC, perturbed angle correlation COMP, composition RBS, Rutherford backscattering spectroscopy EPM, electron probe microanalysis EDAX, energy or wavelength dispersive X-ray, elemental mapping ESCA, electron spectroscopy for chemical analysis AES, Auger electron spectroscopy RI, radioactive isotope analysis SSB, surface sensitive behaviour MH, microhardness COR, corrosion WR, wear TNS, tensile properties HTS, high temperature stability SC, superconductivity measurements MAG, magnetic OPT, optical ANG, results quoted by analytical method not given THRY, theory MT, mass transport HF, heat flow

in more details on the general topic, mechanisms and methodology of laser surface alloying. These are listed at the end of the Bibliography under [106-111].

We hope that this bibliography will serve several purposes. The first is to assemble in one place references of work in this area of research. The manner in which Table I has been constructed allows one to quickly extract the alloying elements, substrates, laser sources, and analyses performed for each reference. It is hoped that the authors will be made aware of both references missed and those forthcoming in order to be able to update the tabulated material presented here.

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